



# **Electrical Resistance Change of Ceramic Matrix Composites in Response to Applied Load and Microstructural Damage**

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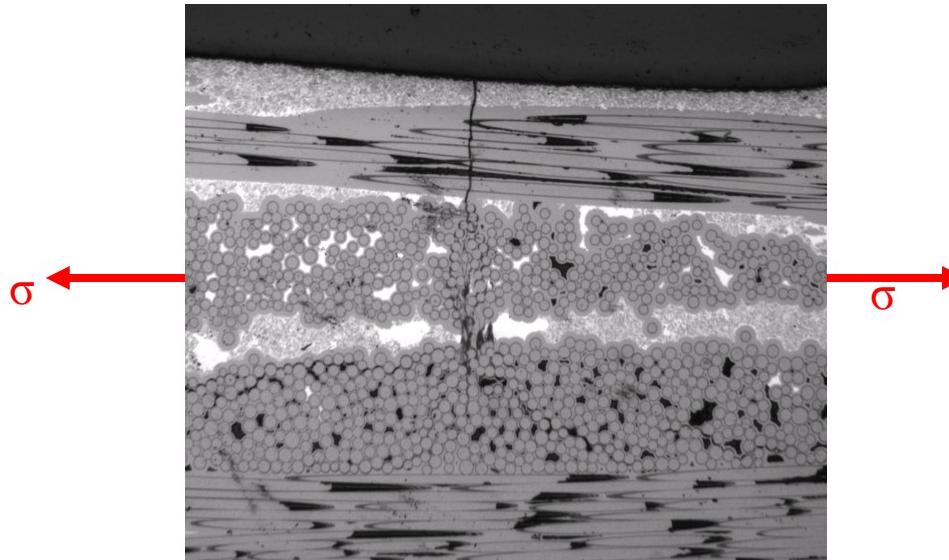
# Why Electrical Resistance?

It would be beneficial to accurately detect small-scale transverse matrix cracks in a CMC coupon or component

As cracks form, only bridging fibers can carry current → resistance increases

For MI SiC/SiC, the matrix is more conductive than the fibers, which should give high sensitivity to crack formation

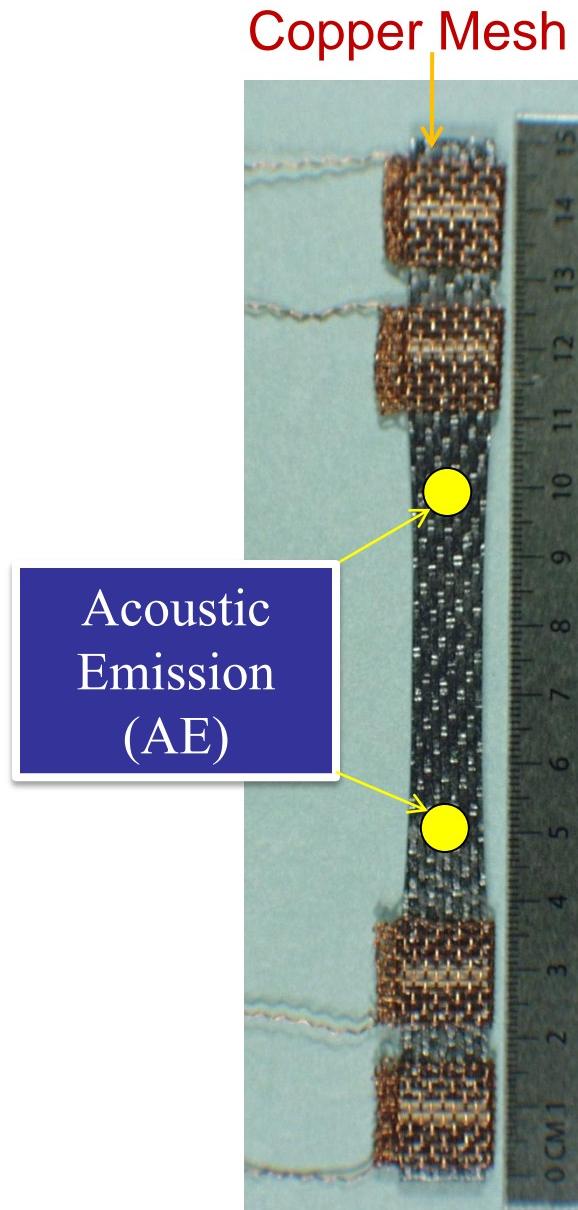
This is a relatively simple technique compared to other inspection methods





# Experimental Procedure

- Resistance measured by four-point probe method
- Silver paint on surface for lower contact resistance
- Grip region wrapped with copper mesh (Used as electrical contact)
  - Offers a simple way of attaching electrical leads for elevated temperature tests
  - For select room-temperature samples, voltage of the contour gage section was also monitored
- Gripped with ceramic wedge grips (for electrical insulation)
- Capacitance strain gage used with 1% range over 25mm
- Resistance monitored with Agilent 34420A micro-Ohm meter
- Acoustic emission monitored by 50kHz to 2MHz sensors just outside the gage section





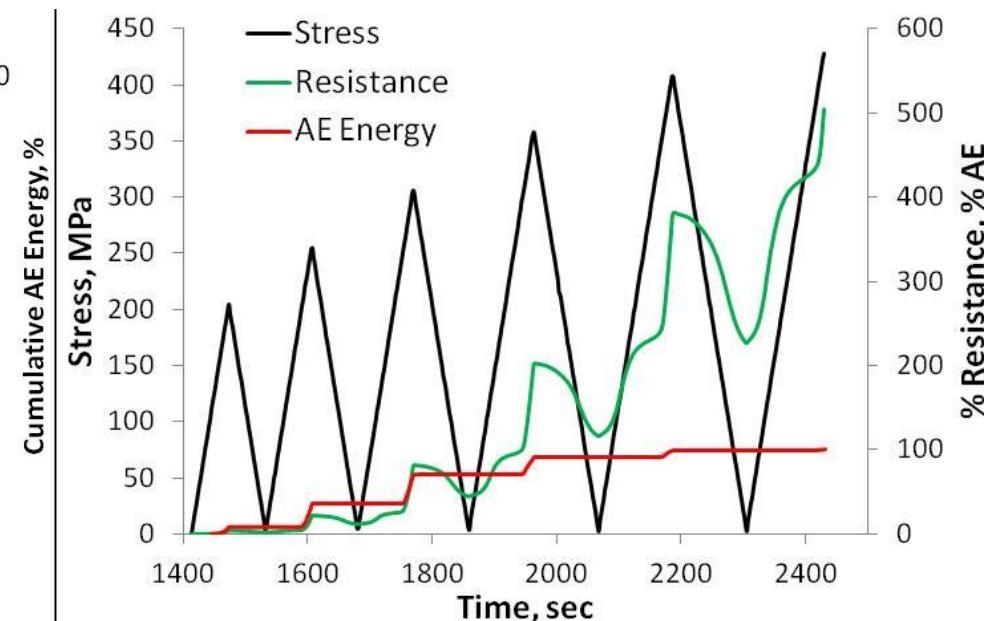
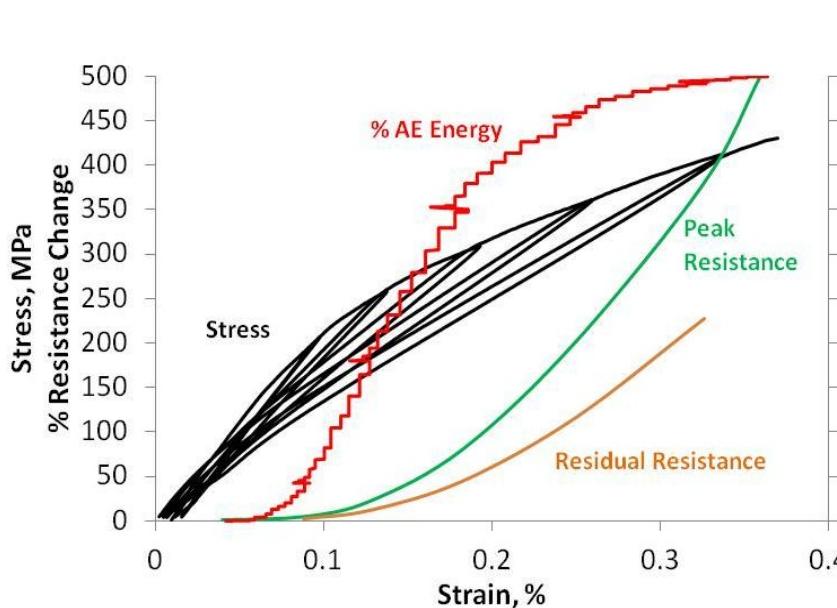
# Room Temperature Damage Characterization

## *Syl-iBN/ Slurry Cast MI Matrix Woven Composite ( $f = 0.38$ )*

Sample	Max Stress, MPa	Stiffness, GPa	Initial Resistivity, $\Omega\text{-cm}$	Resistance change, %	Etched Crack Density, $\text{mm}^{-1}$	PL Stress, MPa	1 <sup>st</sup> Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
1	247*	246	0.027	17.9	0.9	170	138	140
2	300*	237	0.025	78.5	3	180	132	140
3	430	323	0.026	504	10	NA	103	115
4	440	253	0.023	580	--	180	139	110
5	442	260	0.024	466	11.5	170	147	150

Note: all samples have eight plies with a BN interphase, 800 fibers per tow, and total fiber volume fraction of 0.38

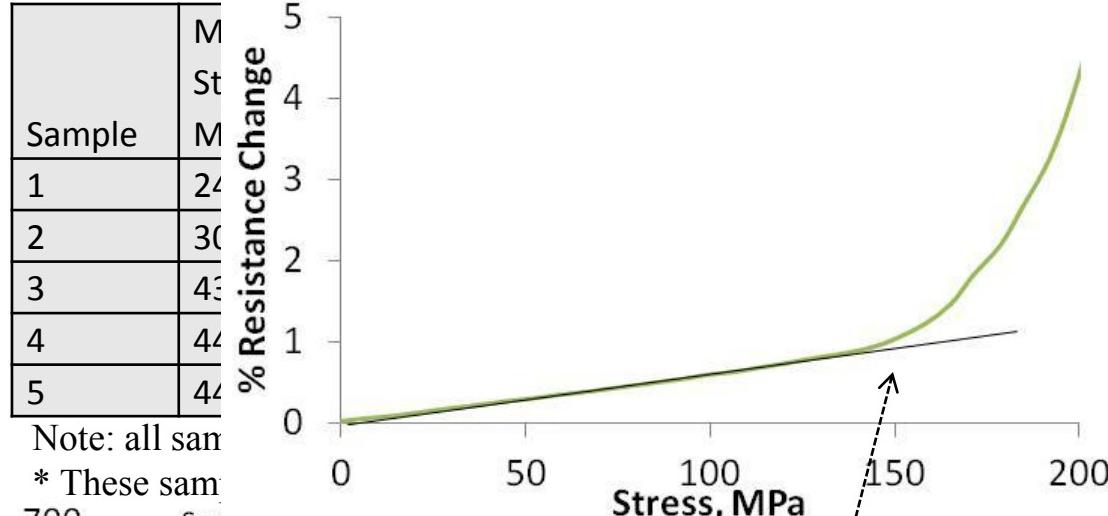
\* These samples were unloaded prior to failure to measure crack density





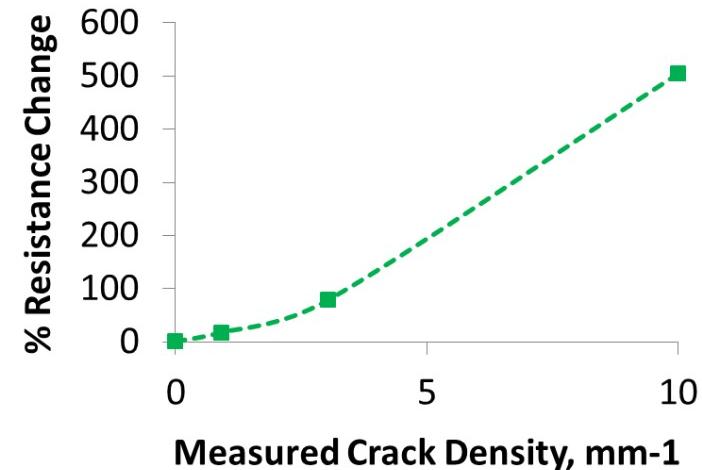
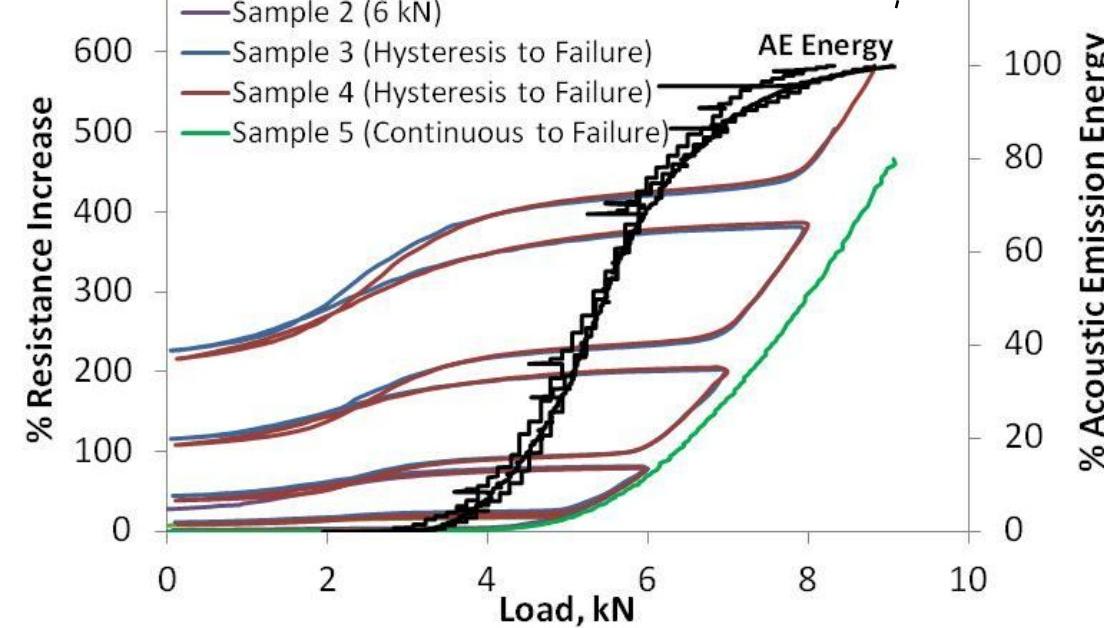
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170	138	140
180	132	140
NA	103	115
180	139	110
170	147	150

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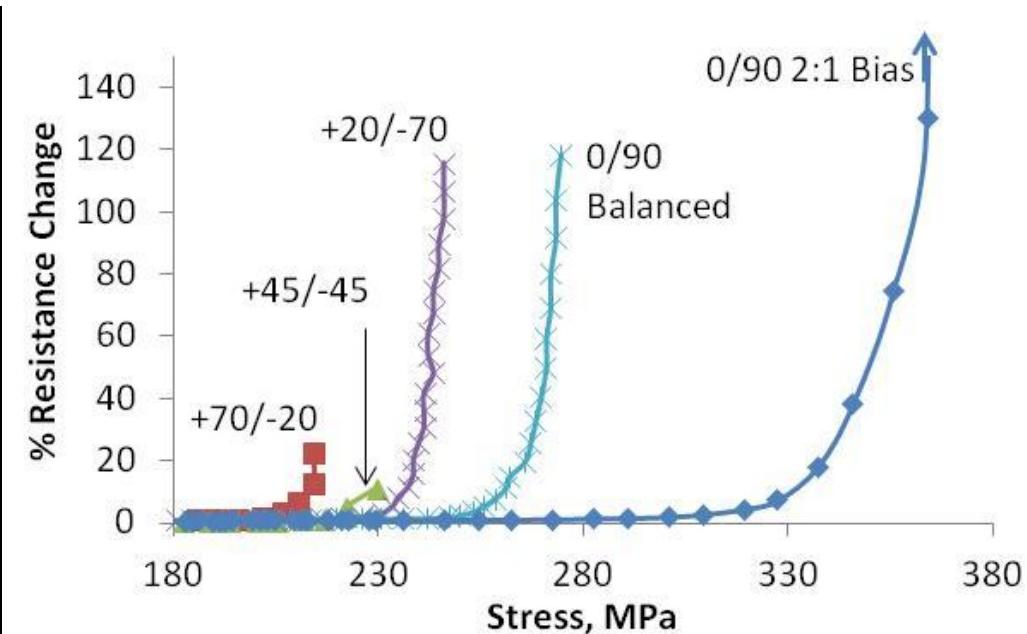
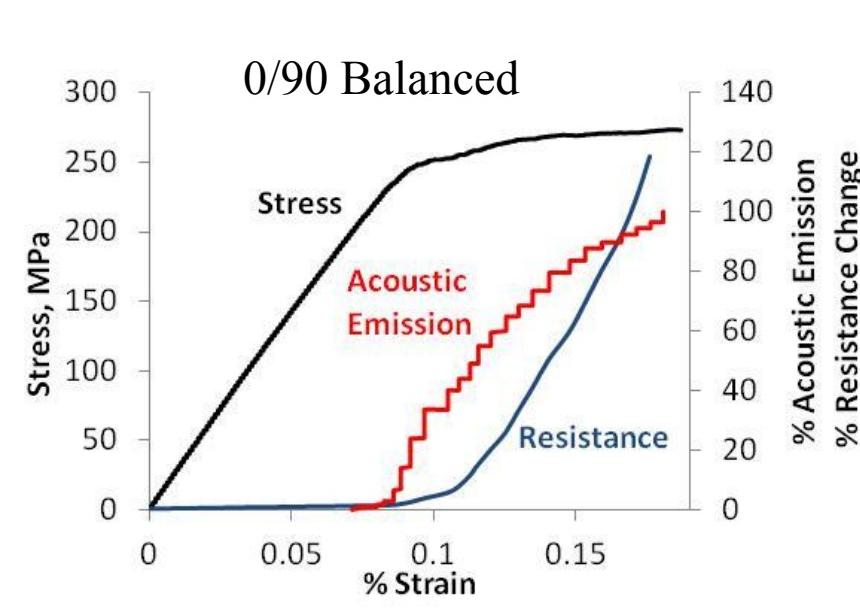


# Room Temperature Damage Characterization

**GE Hi-NicS/ Pre-preg MI Matrix Laminate composite ( $f = 0.22$ )**

Sample	Max Stress, MPa	Stiffness, GPa	Initial Resistivity, $\Omega\text{-cm}$	Resistance change, %	PL Stress, MPa	1 <sup>st</sup> Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
0/90 balanced (0°)	274	288	0.024	118	248	208	230
0/90 balanced (+/-45°)	228	284	0.027	11.2	225	217	215
0/90 balanced (+20/-70°)	235	282	0.026	129	232	204	210
0/90 balanced (+70/-20°)	210	247	0.027	22.4	210	168	160
0/90 biased 2:1 in 0°	300	292	0.027	530	310	262	260

Note: all samples have eight plies with a BN interphase and total fiber volume fraction of 0.22



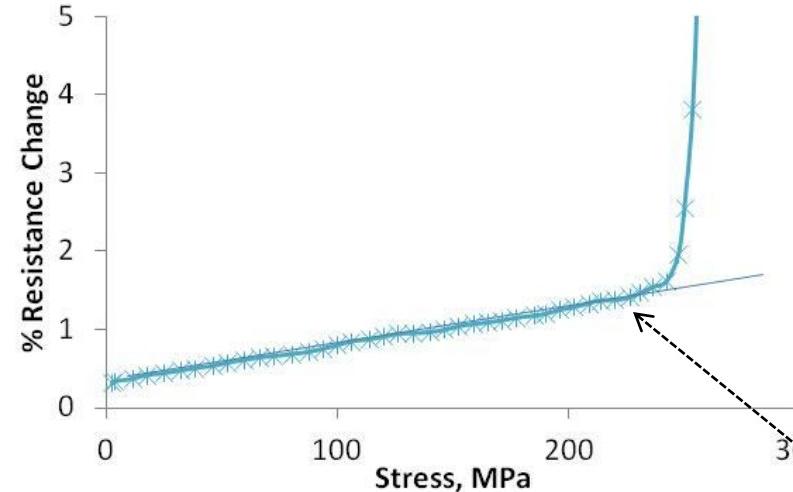


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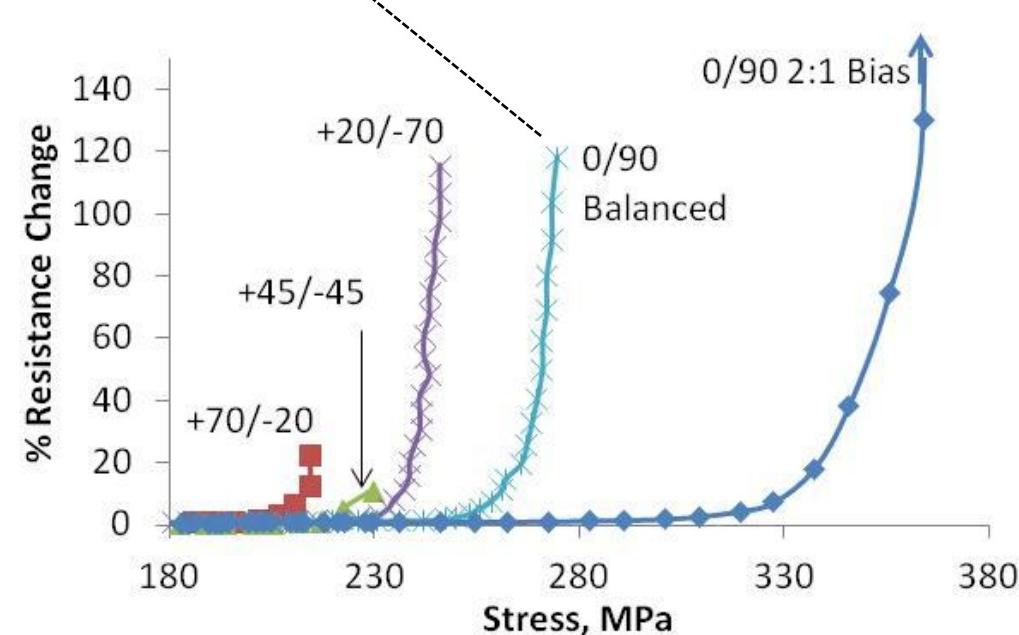
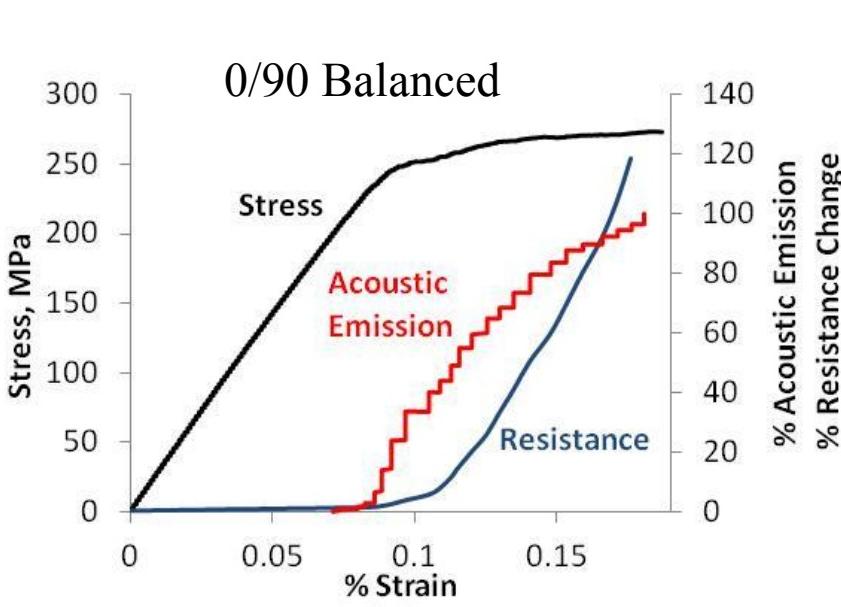
Sample
0/90 balanced (0)
0/90 biased 2:1

Note: all sample



PL Stress, MPa	1 <sup>st</sup> Loud AE Event, MPa	Stress where Resistance is non-linear, MPa
248	208	230
225	217	215
232	204	210
210	168	160
310	262	260

volume fraction of 0.22





# The Need for a Model

## Key Benefits

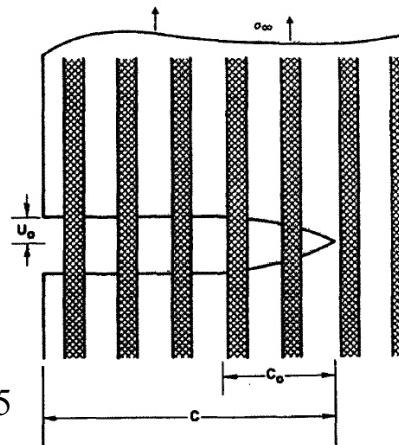
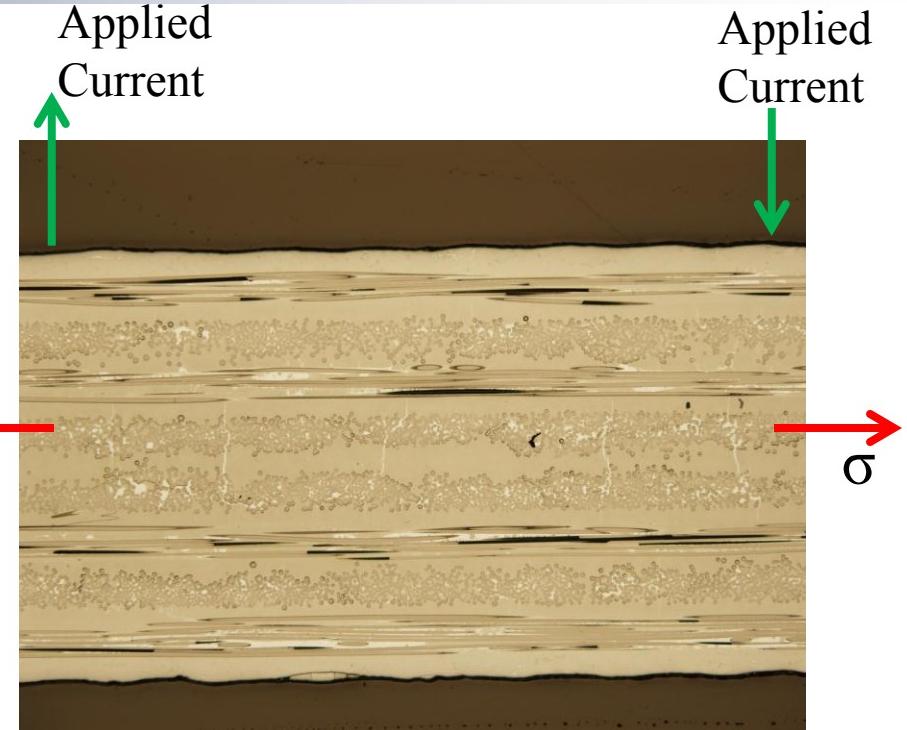
- Can be measured in-situ
- Resistance changes permanently for inspection at zero load
- The deviation from linearity correlates with the proportional limit stress
- Resistance is sensitive to crack formation
- Can be used at elevated temperature
- Repeatable

## Key Concerns

- The data needs to be related to the microstructural changes in the material
- Many variables at room temperature
  - crack formation, crack growth, fiber/interphase sliding, fiber breaks
- Even more variables at elevated temp
  - creep of constituents, oxidation, change in resistivity with temperature
- Need to understand what effect each mechanism has on resistance

# Model Development

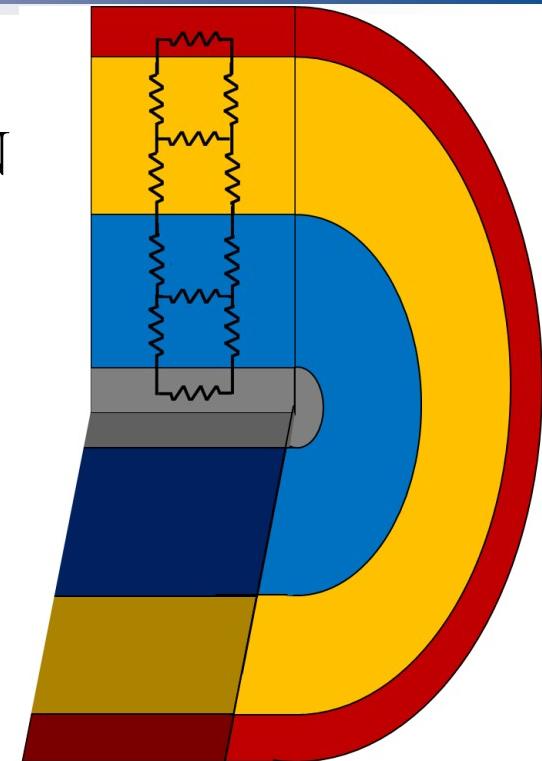
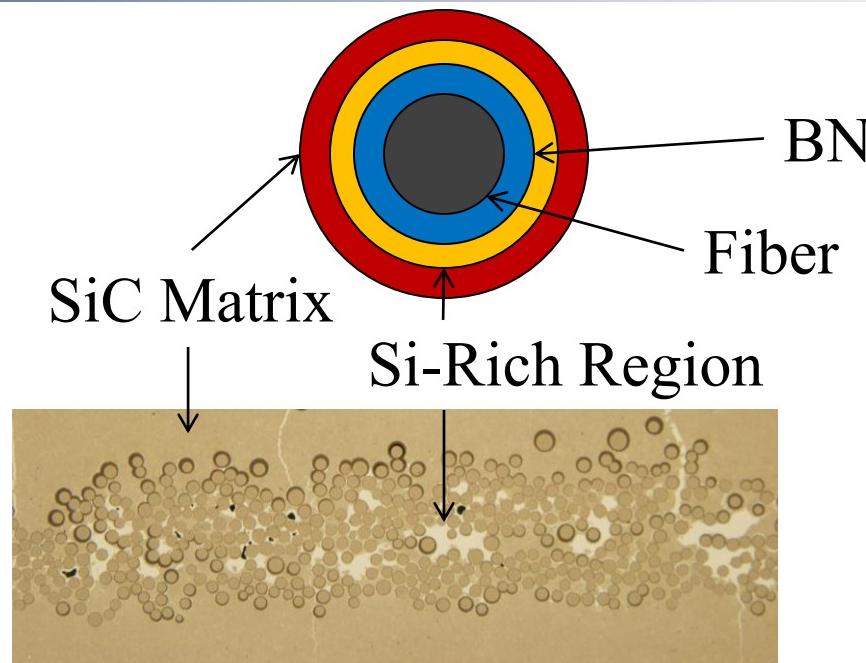
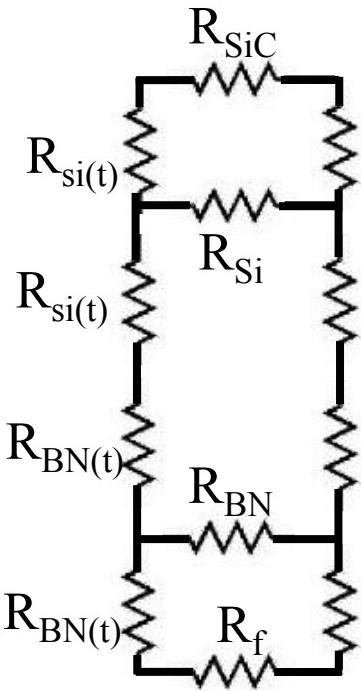
- Current is applied at the MI matrix surface
- The matrix conductivity is an order of magnitude greater than the fibers (MI contains Si)
- Also, BN insulates the fibers from the matrix
- When the matrix is cracked, current must be transferred to the fibers
- The question is how?



Marshall, Cox, Evans 1985



# Unit Cell for GE pre-preg MI



- The composite is treated as a series of concentric cylinders
- Fibers are surrounded by BN, then a Si rich region in the ply, then the bulk MI
- All the fibers are treated as one, while maintaining the same relative volume fractions for constituents
- $90^\circ$  plies are neglected in this case, since the fibers will not bridge the cracks and they do not provide a continuous path (BN interphases in series)
- Each unit cell represents a specified length along the loading direction ( $10 \mu\text{m}$ )



# Unit Cell for GE pre-preg MI

- Resistance along the length of each constituent:

$$R_c = \frac{\rho A_c}{L}$$

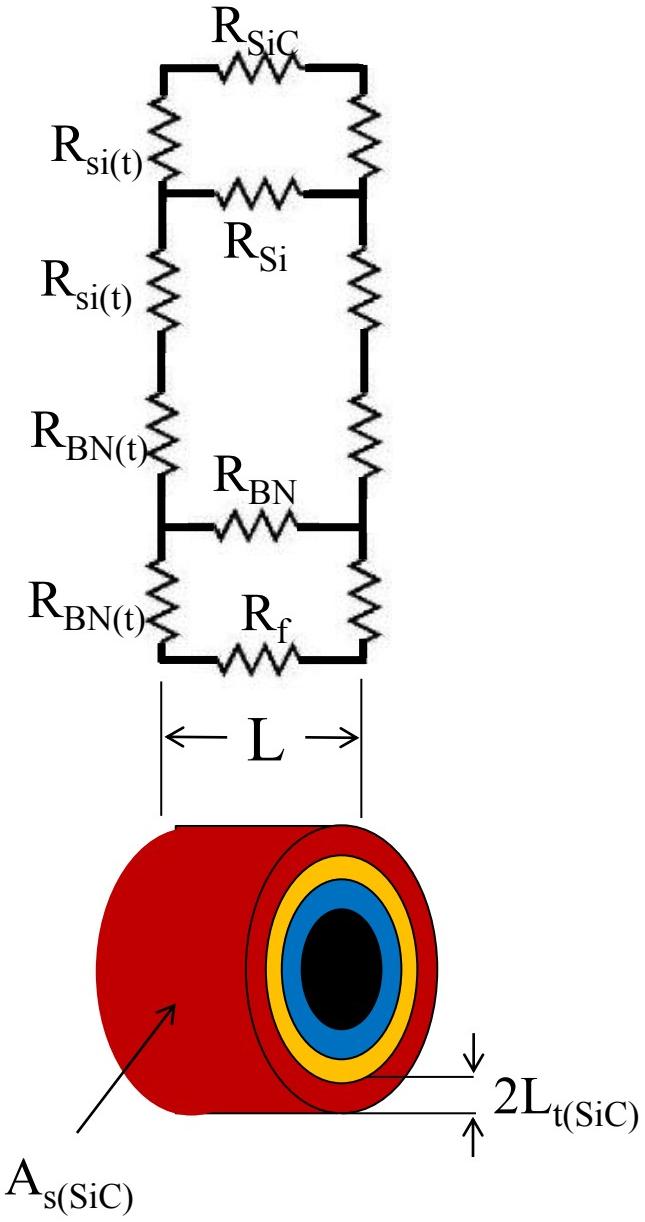
$\rho$  is the constituent resistivity  
 $A_c$  is the cross-sectional area  
L is the length of the unit cell

- Similar equations describe the resistance of the constituents in the transverse (radial) direction:

$$R_t = \frac{\rho A_s}{L}$$

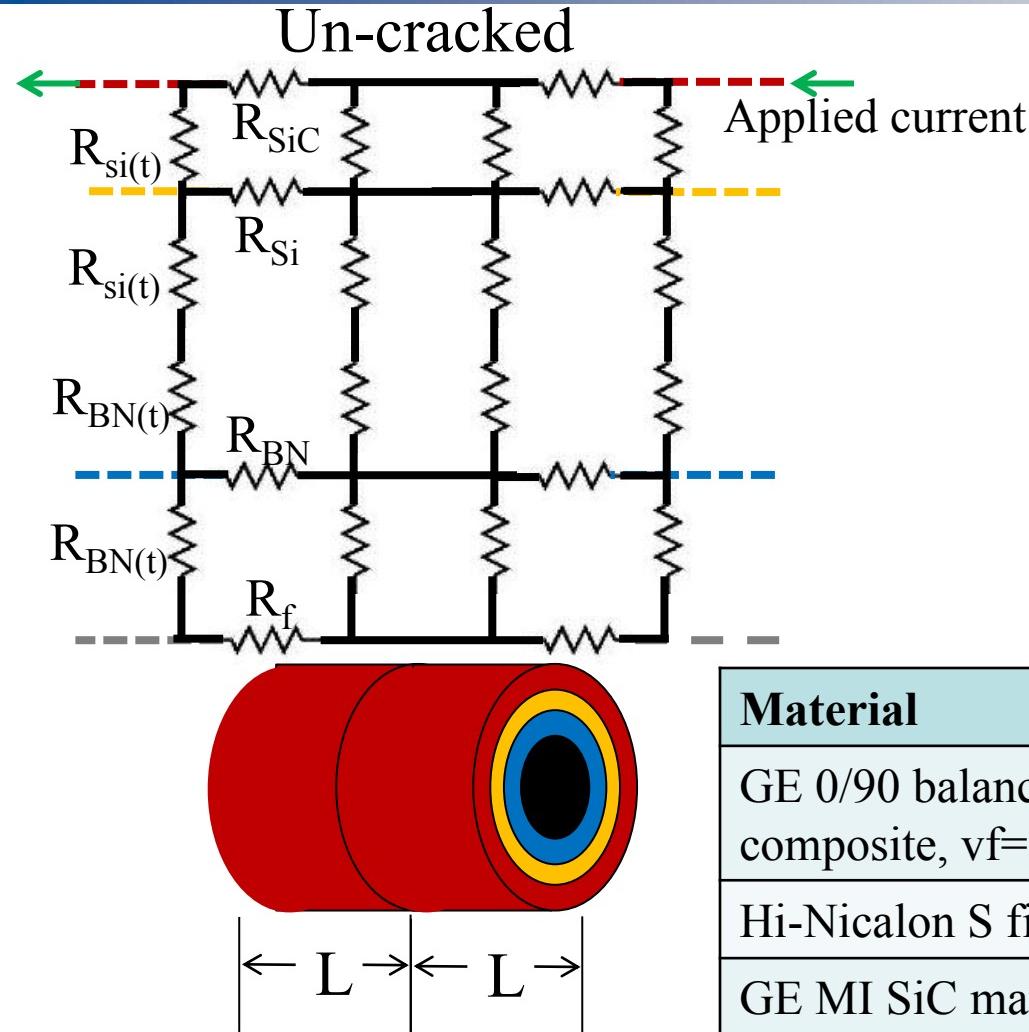
$A_s$  is  $\frac{1}{2}$  of the mean surface area of the cylinder  
 $L_t$  is  $\frac{1}{2}$  of the thickness of the cylinder

Material	Resistivity, $\Omega\text{-cm}$
GE 0/90 balanced composite, vf= 0.22	0.027 (measured directly)
Hi-Nicalon S fiber	0.1 (from literature)
GE MI SiC matrix	0.037 (measured directly)
Si rich region in ply	Initially Unknown- very low
BN interphase	Initially Unknown- very high





# Model Parameters

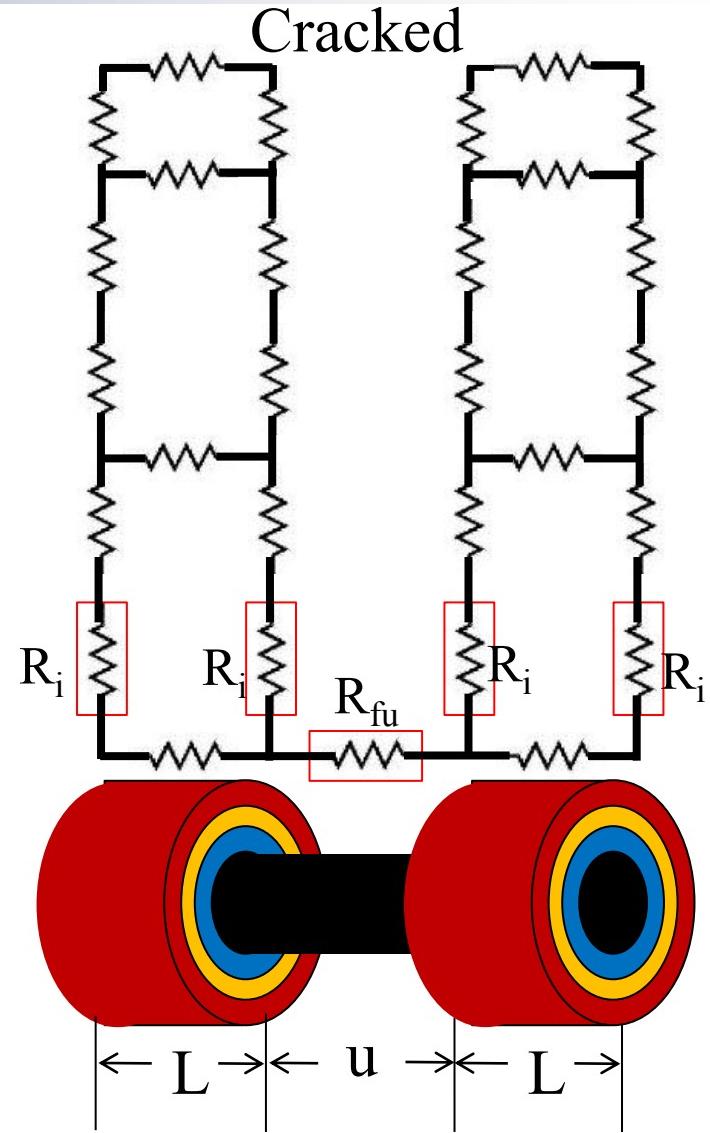
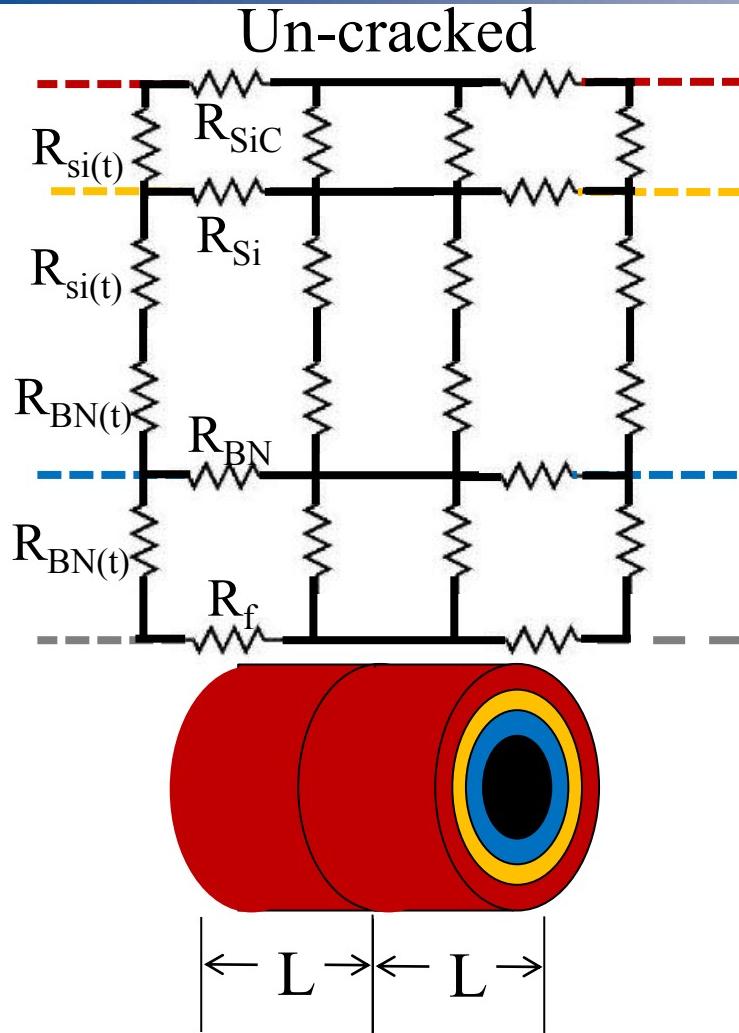


- Current is applied to the first and last cells at the outer SiC layer
- Adjusting the BN resistivity has no effect on the un-cracked composite since the matrix resistivity is low
- Resistivity of the Si-rich region can be adjusted to fit the initial composite resistivity

Material	Resistivity, $\Omega\text{-cm}$
GE 0/90 balanced composite, $vf = 0.22$	0.027 (measured directly)
Hi-Nicalon S fiber	0.1 (from literature)
GE MI SiC matrix	0.037 (measured directly)
Si rich region in ply	0.0038 (from model)
BN interphase	Initially Unknown- very high



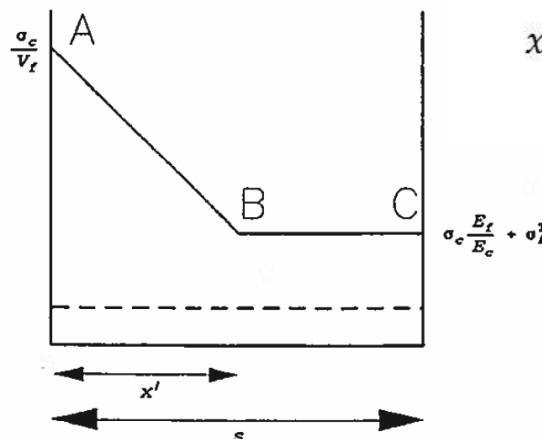
# Model Description





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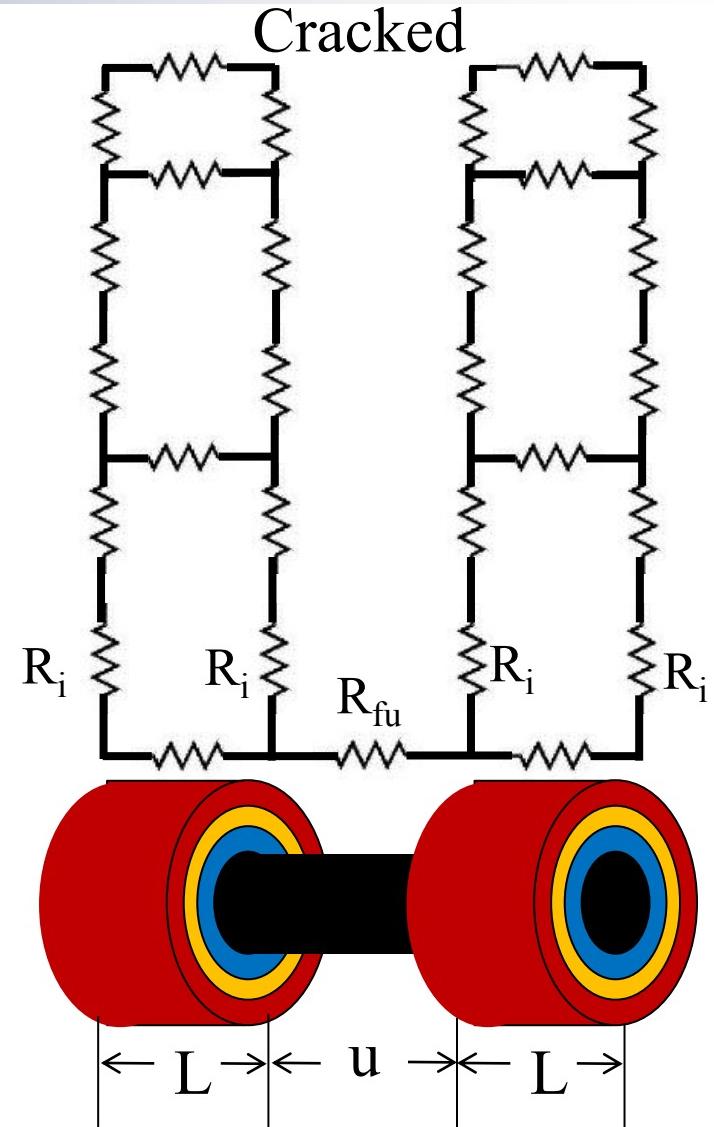
- The matrix and BN circuits are broken in the crack
- The fiber will de-bond from the BN and slide over a distance  $x^1$
- An interfacial resistance is introduced between the BN and fiber
- The surrounding unit cells within the distance  $x^1$  will all be affected



8 Stress profile in the fibres for a cracked laminate at an applied stress  $\sigma_c$

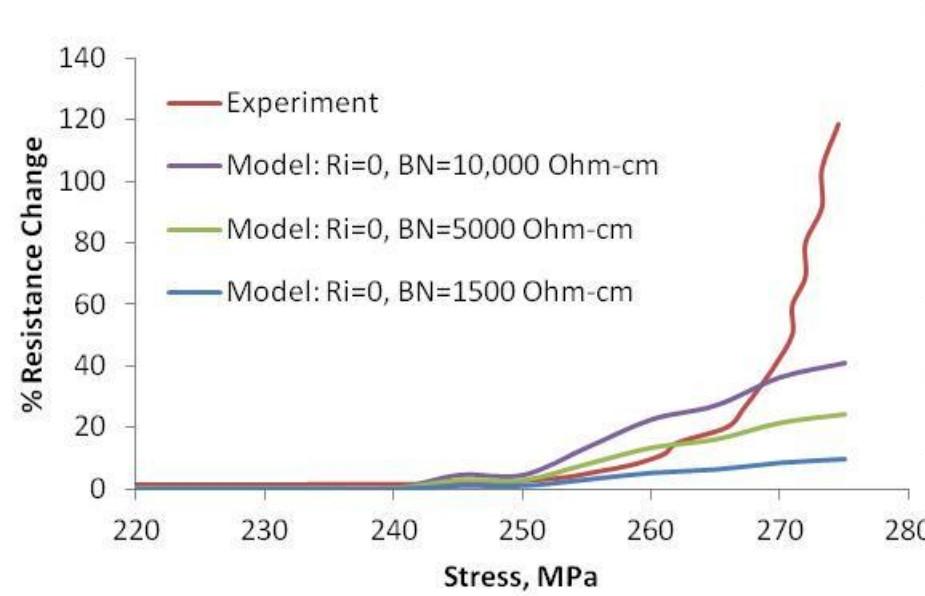
$$u = \frac{\sigma^2 R}{4\tau v_f^2 E_f \left( 1 + \frac{E_f v_f}{E_m (1 - v_f)} \right)}$$

Marshall, Cox, Evans 1985



# Model Calibration

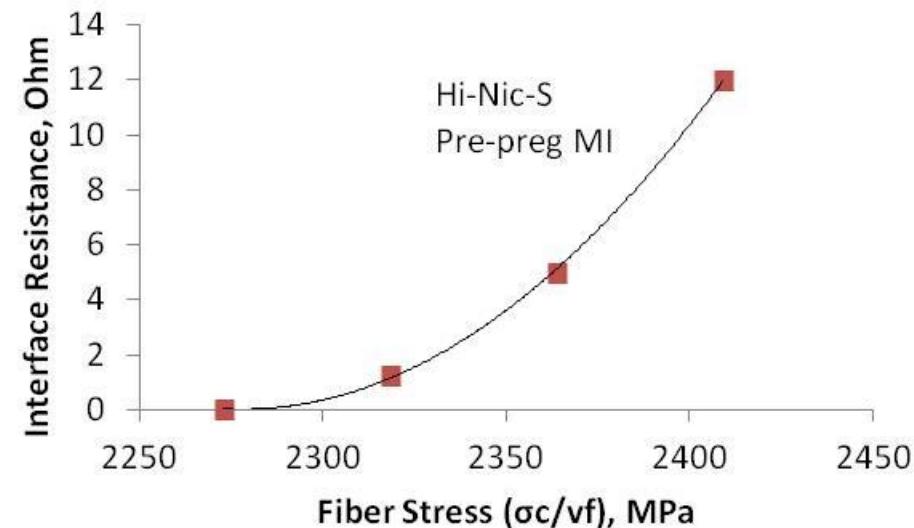
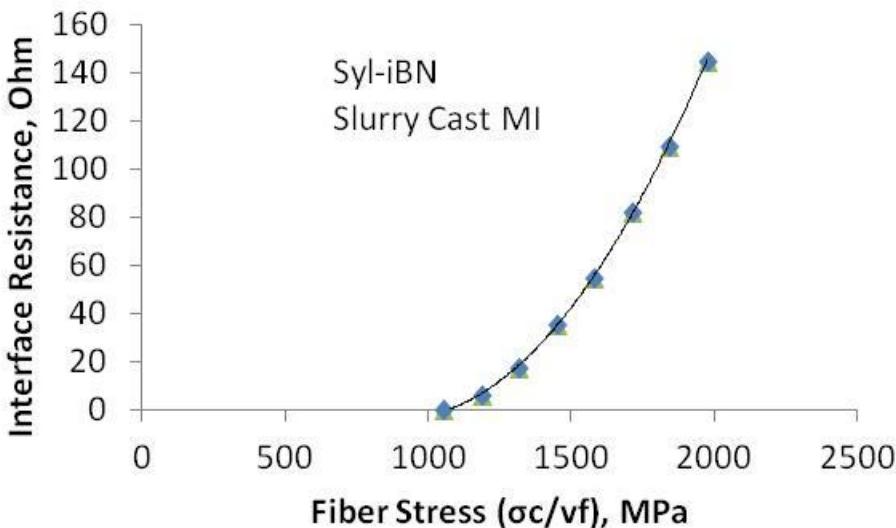
- Acoustic emission energy is used as an estimate of crack density as a function of stress
- The model introduces cracks at incremental stress levels, according to the AE data
- The cracks are randomly distributed and the overlap length can be specified
- The resistivity of BN is determined by setting  $R_i=0$  and adjusting the BN
- If the value is too high, the model will be too sensitive to cracks
- The maximum BN resistivity was chosen such that the model would never overshoot the experiment in the extreme case of  $R_i=0$



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GE 0/90 balanced composite, $vf= 0.22$	0.027 (measured directly)
Hi-Nicalon S fiber	0.1 (from literature)
GE MI SiC matrix	0.037 (measured directly)
Si rich region in ply	0.0038 (from model)
BN interphase	1500 (from model)

# Model Results

- Interfacial resistance  $R_i$  is assumed to be uniform over the slip length  $x^1$ , but changing with stress
- It is expected that the interfacial resistance increases due to relative sliding between fiber and matrix
- We see that  $R_i$  must be proportional to  $\sigma^2$  to fit the experimental data

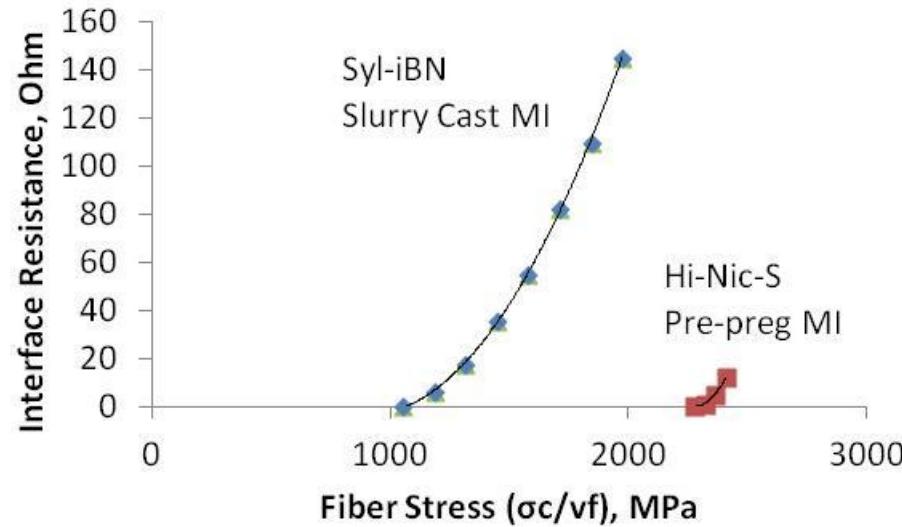


- Sliding distance of the fiber at the crack surface is proportional to  $\sigma^2$

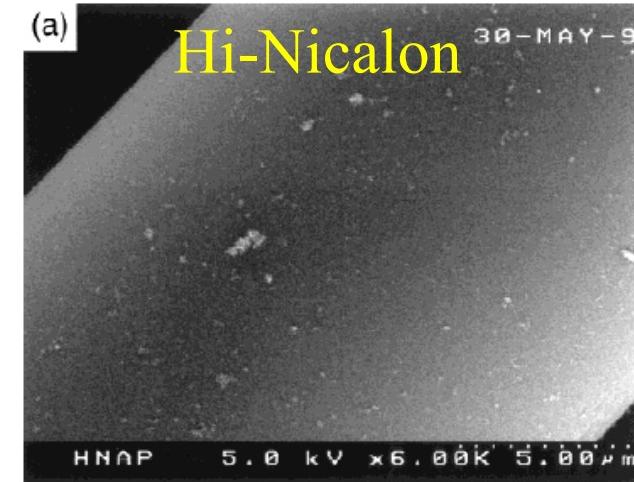
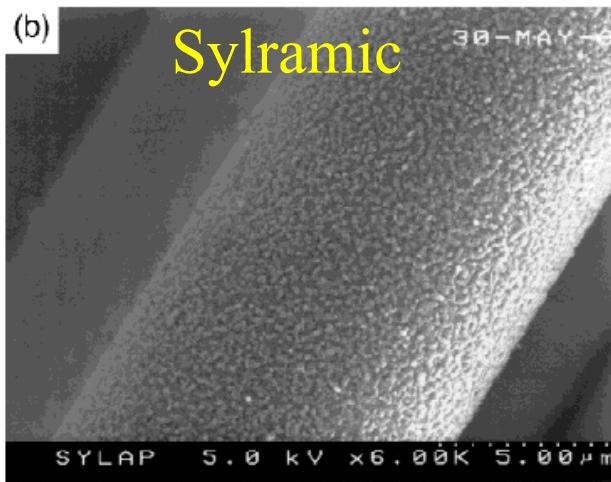
$$u = \frac{\sigma^2 R}{4\tau v_f^2 E_f \left( 1 + \frac{E_f v_f}{E_m (1 - v_f)} \right)}$$

Marshall, Cox, Evans 1985

# Model Results

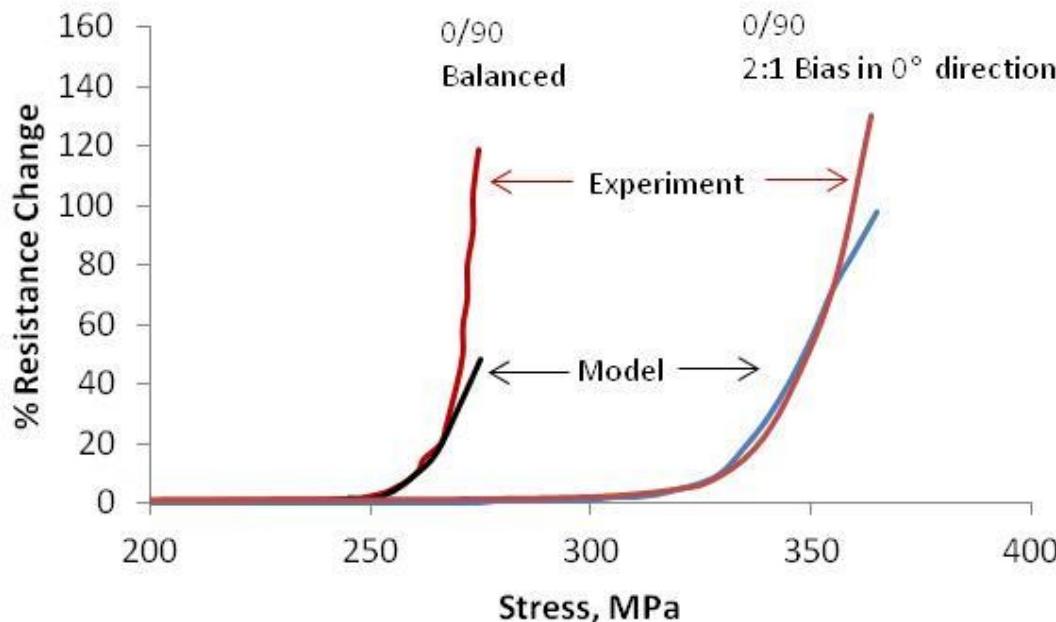


- The difference in magnitude of  $R_i$  for the two composites is likely due to fiber roughness



# Model Results

- By using the parabolic relationship for  $R_i$ , the model fits the experiment for the balanced 0/90 GE pre-preg composite at 98% of the strength
- The same parabolic function generated from the balanced composite was used to model an unbalanced 0/90 sample with 2:1 bias in the loading direction
- The model also fit this data, which indicates that the relationship is more than a mere curve fit





## Conclusion

- Electrical resistance offers a way of monitoring damage in CMC's
- Several factors influence the electrical properties
- A discrete model has been developed to understand the mechanisms causing electromechanical changes
- The model verifies that the interfacial resistance is a function of stress squared, consistent with fiber sliding



# Future Work

Examine the effect of the following:

- $R_i$  varying along the slip length
- Fiber breaks
- Load-unload- reload
- Varying cross-section along the sample
- Temperature
- Creep
- Environment